Linking dynamic and thermodynamic properties of cuprates: An angle-resolved photoemission study of \((\text{Ca}_{x}\text{La}_{1-x})(\text{Ba}_{1.75-x}\text{La}_{0.25+x})\text{Cu}_3\text{O}_y\) \((x = 0.1\) and \(0.4)\)

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(Received 25 August 2013; published 31 March 2014)

We report angle-resolved photoemission spectroscopy on two families of high-temperature superconductors \((\text{Ca}_{x}\text{La}_{1-x})(\text{Ba}_{1.75-x}\text{La}_{0.25+x})\text{Cu}_3\text{O}_y\) with \(x = 0.1\) \((T_c^{\text{max}} = 56\) K) and \(x = 0.4\) \((T_c^{\text{max}} = 82\) K). The Fermi surface (FS) is found to be independent of \(x\) or \(y\), and its size indicates extreme sample-surface overdoping. This universal FS allows the comparison of dynamical properties between superconductors of similar structure and identical doping, but different \(T_c^{\text{max}}\). We find that the high-energy \(|E| > 50\) meV) nodal velocity in the \(x = 0.4\) family is higher than in the \(x = 0.1\) family. The implied correlation between \(T_c^{\text{max}}\) and the hopping rate \(t\) supports the notion of kinetic energy driven superconductivity in the cuprates. We also find that the antinodal gap is higher for the \(x = 0.4\) family.

DOI: 10.1103/PhysRevB.89.121119

PACS number(s): 74.72.Gh, 74.25.Jb, 79.60.−i

The recent synthesis of charge compensated \((\text{Ca}_{x}\text{La}_{1-x})(\text{Ba}_{1.75-x}\text{La}_{0.25+x})\text{Cu}_3\text{O}_y\) (CLBLCO) single crystals, with \(x = 0.1\) and \(x = 0.4\), facilitates an investigation of the relationship between their dynamical properties, such as the electronic dispersion relation \(E(k)\) and their thermodynamic property \(T_c\), while applying subtle crystal structure changes [1]. Since the valence of Ca and Ba is equal, \(x\) has a minute effect on crystal structure but a large effect on \(T_c\). Therefore, CLBLCO allows experiments where the correlations between \(T_c\) and a single parameter are explored. Experiments of such nature can reveal the mechanism for cuprates’ superconductivity. In the present work, we measure the electron dispersion \(E(k)\) of two extreme samples of CLBLCO crystals, using angle-resolved photoemission spectroscopy (ARPES), and look for correlations between properties of \(E(k)\) and \(T_c\). In particular, we focus on the nodal velocity. Previously, similar studies could only be done by comparing cuprates with very different structures and levels of disorder [2].

CLBLCO is similar to YBCO in crystal structure, but has no oxygen chain ordering and is tetragonal for all \(x\) and \(y\) [3]. This simplifies the ARPES interpretation. While \(x\) alters the calcium-to-barium ratio, the lanthanum content in the chemical formula remains constant. We define four CLBLCO “families” as samples with different \(x\), namely, \(x = 0.1, 0.2, 0.3, 0.4\). The parameter \(y\) signifies the oxygen level, which drives the system between different phases. By varying \(x\) and \(y\) in the chemical formula, one can generate phase diagrams that are similar in shape yet differ in the maximum of \(T_c\), \(T_g\), and \(T_N\), and in the critical oxygen level at which the nature of the phase diagram changes. The phase diagram is presented in Fig. 1(a) [4]. It is worth noting that the only structural properties that vary with \(x\) or \(y\) are the Cu-O-Cu buckling angle, bond length, and CuO\(_2\) plane doping efficiency \(K(x)\). The crystallographic parameters were measured with powder neutron diffraction [5]. The buckling angle decreases by 0.5 degrees as \(x\) increases between families. The bond length varies from 3.88 Å for \(x = 0.4\) to 3.91 Å at \(x = 0.1\). The doping efficiency is determined by in-plane \(^{13}\)O nuclear quadrupole resonance (NQR) [4]. The variation in the number of holes on an oxygen \(\Delta n_p\) is given by \(\Delta n_p = K(x)(y - y_N)\), where \(y_N\) is defined as the doping at which \(T_N\) starts to drop [see Fig. 1(a)] [4].

The superexchange parameter \(J\) for each CLBLCO family was previously determined with muon spin rotation (\(\mu\)SR) (magnetization) versus temperature measurements [6] and with two-magnon Raman scattering [7]. Figure 1(b) depicts the superexchange \(J\) and glass temperature \(T_g\) (both from \(\mu\)SR), and \(T_c\), all normalized by \(T_c^{\text{max}}\), as a function of \(\Delta n_p\). A universal phase diagram appears, demonstrating that \(T_c^{\text{max}}\) scales like \(J\) [4], which implies that \(T_c^{\text{max}}\) is determined by the overlap of the orbital occupied by electrons on neighboring sites. Orbital overlaps also determine the hopping parameter \(t\), and the scaling of \(T_c^{\text{max}}\) with \(J\) meaning that kinetic energy controls the superconducting phase transition. However, \(J\) is determined in the AFM phase, which is “far,” in terms of doping, from the superconducting phase. A question arises: are the orbital overlaps important in the superconducting phase as well? In this phase \(t\) can be measured directly. Here, we extract \(t\) from \(E(k)\) as the velocity in the nodal direction. We find correlations between \(T_c^{\text{max}}\) and \(t\), and confirm the famous relation \(J \propto t^2\) [8]. This suggests that the band structure is rigid as a function of doping, as suggested by recent resonance inelastic x-ray scattering experiments [9]. By the same token, we also measure the antinodal gaps and compare them with Hamiltonian parameters.

The ARPES experiments were performed on the SIS beamline at the Swiss Light Source on CLBLCO single crystals. These unique crystals were grown using the traveling floating zone method. A detailed discussion about growth and characterization of these crystals is given in [1]. For this experiment, samples with \(x = 0.1\) and \(x = 0.4\) were used. The samples were mounted on a copper holder with silver glue to improve electrical conductivity. The Fermi level and resolution were determined from the polycrystalline copper sample holder. The samples were cleaved in situ using a glued-on pin at \(T = 10–20\) K. Circularly polarized light with \(h\nu = 50\) eV was used. The spectra were acquired with a VG Scienta R4000 electron analyzer. Despite a base pressure of \(5 \times 10^{-11}\) torr, the samples’ surface lifetime was only a few hours and a high-intensity beam was required for quick...
FIG. 1. (Color online) (a) The phase diagram of CLBLCO showing the Néel ($T_N$), glass ($T_g$), and superconducting ($T_c$) temperatures over the full doping range for the four families. $y_N$ indicates the oxygen level where $T_N$ start to drop. (b) The unified phase diagram of CLBLCO. The critical temperatures, and $J$ extracted from $T_N$, are divided by $T^{\text{max}}_c$ and plotted as a function of doping variation in the oxygen orbital $\Delta n_o$. (c)–(e) Raw ARPES date measured on a CLBLCO $x = 0.1$ crystal at $T = 16$ K. The numbers on the figures correspond to cut trajectories illustrated in the inset of the phase diagram. (f)–(h) The same as (c)–(e) but for a sample with $x = 0.4$ measured at $T = 11$ K.

As a consequence, the energy resolution in our experimental conditions was limited to 17–22 meV.

In Fig. 1, we present ARPES data collected from CLBLCO for the two samples: $x = 0.1$ is presented in panels (c)–(e), while $x = 0.4$ is depicted in panels (f)–(h). The data were collected at $T = 16$ K and 11 K for $x = 0.1$ and 0.4, respectively. All spectra are normalized by the measured detector efficiency. For each sample, intensities along three cuts are shown. The cuts are illustrated and numbered on the Fermi surface (FS) drawing in the inset of Fig. 1(b). Cuts numbered 1 and 2 are along $k_x$ ($\Gamma$-M direction). These cuts allow better sensitivity to the gap size at the antinode. Cut number 3 is along the diagonal line of the BZ (\Gamma Y). In this configuration, a measurement of velocity in the nodal direction is possible. The number on the bottom of each ARPES panel indicates the cut from which data are collected.

In Figs. 1(c) and 1(f), spectra near the antinode are plotted. While $x = 0.1$ shows high-intensity spectra up to $E_f$ where no gap is visible, the $x = 0.4$ sample shows a depletion of intensity close to $E_f$, indicating a gap in the spectra at the antinode. For the $x = 0.1$, the gap, if one exists, is smaller than the experimental resolution. In Figs. 1(d) and 1(g), we plot the intensity closer to the node. For both the $x = 0.1$ and 0.4 sample, we clearly see the spectra crossing $E_f$ indicating a closed gap in the nodal region. Finally, in Figs. 1(e) and 1(h), both nodal cuts are seen, and again, the spectra cross $E_f$, indicating an absence of a gap along the Fermi arc for both samples. The last panels also show a clear dispersion from which the nodal velocity is extracted.

In Fig. 2, we show the FS in the first Brillouin zone (BZ), for the two CLBLCO samples: $x = 0.1$ [Fig. 2(a)] and 0.4 [Fig. 2(b)]. The FS was obtained by integrating 10 meV around the chemical potential. The ARPES intensity is displayed in a false-color scale as a function of $k_x$ and $k_y$. By comparing the shape of the FS, we can see that the $x = 0.4$ sample exhibits a Fermi arc structure [10], which is typical for an antinodal gap. As for the $x = 0.1$ sample, the arc is not present, and we observed strong intensity at the antinode, comparable to the intensity near the nodal region. Unlike previously reported FS measurements of YBCO [11], there is no apparent chainlike structure in the CLBLCO FS, as expected. The red line is a fit to a tight binding (TB) model up to three nearest neighbors hopping. The fit parameters will be discussed below. The fit for both FSs gives the same size, as can be seen in Fig. 2. In fact, the FS of a variety of samples was measured and found to be identical regardless of family ($x$) or bulk oxygen level ($y$).

A clearer comparison of the FS size and doping between families can be obtained by examining the node-to-node and antinodal distances. In Fig. 2(c), we show momentum distribution curves (MDCs) at zero binding energy ($E_f$) measured in a nodal cut (“cut 3”), for both samples. The MDC for $x = 0.4$ is sharper than for $x = 0.1$, but the peak-to-peak distance is equal for both MDCs. Similarly, Fig. 2(d) depicts an MDC measured in the antinode (“cut 2”) at $E_f$. Here, the MDC of $x = 0.1$ is clearer than that of $x = 0.4$ because of an open gap, but again the peak separation for both samples is identical.
antinodal cut, for the red, Fig. 1(h) samples. (d) MDCs at zero binding energy along the

Close to the node, we observe a strong peak at zero energy

at

The data were obtained at \( T = 16 \) K for the \( x = 0.1 \) sample and at \( T = 11 \) K for the \( x = 0.4 \) sample. Both samples were prepared with optimal doping and verified with a SQUID magnetometer. The red curve is the FS of CLBLCO obtained from tight binding fits to experimental data (see text). (c) MDCs at zero binding energy along a nodal cut, for the \( x = 0.1 \) [black, Fig. 1(e)] and \( x = 0.4 \) [red, Fig. 1(b)] samples. (d) MDCs at zero binding energy along the antinodal cut, for the \( x = 0.1 \) [black, Fig. 1(d)] and \( x = 0.4 \) [red, Fig. 1(g)] samples.

We suspect that the bulk doping independence of the FS is due to the sample being cleaved on a charged plane, inducing surface charge reconstruction. Such behavior was previously reported from measurements of YBCO [11]. From inducing surface charge reconstruction. Such behavior was is due to the sample being cleaved on a charged plane,

the measured nodal peak-to-peak distance as a function of temperature of \( T \)

"cut 2" for the \( x = 0.1 \) sample and \( x = 0.4 \) samples. Thus, we can conclude that the surface doping level of both samples is equal within the experimental error, and that the result is consistent with calculations based on the FS volume.

To investigate the momentum dependence of the gap, we measured the dispersion along \( \Gamma - M \) cuts between “cut 1” and “cut 2” for the \( x = 0.1 \) and \( x = 0.4 \) samples at a cold-finger temperature of \( T = 16 \) K and \( T = 11 \) K, respectively. In Fig. 3, we plot symmetrized EDCs at \( k_f \) as a function of FS angle \( \phi \) (defined in the inset of Fig 1). For the \( x = 0.4 \) sample [Fig. 3(a)], one can see a zero-energy intensity peak close to the node (\( \phi = 36 \)). In contrast, at an angle of \( \phi = 20 \) and lower, we observe an opening of a gap, which grows up to \( \Delta_0 = 40 \) meV at the antinode (\( \phi = 0 \)). The angular dependence of the gap is shown in Fig. 1 of the Supplemental Material [12]. The gap value at the antinode is similar to optimally doped Bi2212 [13,14] and YBCO [15,16].

For the \( x = 0.1 \) sample [Fig. 3(b)], the situation is different. Close to the node, we observe a strong peak at zero energy (\( \phi = 35 \)). As we move to the antinode, the intensity at zero energy is partly suppressed, but unlike the \( x = 0.4 \) sample, there is no full depletion of spectral density at \( E = 0 \). This indicates that a gap is not present in the \( x = 0.1 \) sample, or that it is smaller than the experimental resolution (20 meV). A closed gap was measured with the same resolution for two more \( x = 0.1 \) samples. Thus, we can safely say that \( \Delta_0(x = 0.4) > \Delta_0(x = 0.1) \).

Last but not least, we compare the nodal velocity between families. This study was performed on six \( x = 0.1 \) and seven \( x = 0.4 \) crystals. The dispersion in the nodal direction, previously described in Figs. 1(e) and 1(b), was measured for each sample in two branches with high statistics. The samples were first oriented using a manipulator with six degrees of freedom in a procedure which is described in the Supplemental Material [17]. This lead to a \( k \) resolution of 0.005 (\( \pi/a \)) determined by our ability to find the node to node direction. The peak positions in the MDC of each measured dispersion was extracted and plotted as a function of binding energy. Exemplary dispersions of two samples are shown in Fig. 4. An axis breaker is used in order to show the two branches. The breaker emphasizes the differences between \( k_f \) of the two samples, which in fact is very small. Two different linear regimes are observed. The first regime involves low energies close to \( E_f \), between \(-50 < E < 0 \) meV. The second regime corresponds to high energies where

FIG. 2. (Color online) (a), (b) Spectral weight map in \( k \) space at \( E_f \) (FS) in CLBLCO \( x = 0.1 \) and \( x = 0.4 \) sample, respectively. The data were obtained at \( T = 16 \) K for the \( x = 0.1 \) sample and at \( T = 11 \) K for the \( x = 0.4 \) sample. Both samples were prepared with optimal doping and verified with a SQUID magnetometer. The red curve is the FS of CLBLCO obtained from tight binding fits to experimental data (see text). (c) MDCs at zero binding energy along a nodal cut, for the \( x = 0.1 \) [black, Fig. 1(e)] and \( x = 0.4 \) [red, Fig. 1(b)] samples. (d) MDCs at zero binding energy along the antinodal cut, for the \( x = 0.1 \) [black, Fig. 1(d)] and \( x = 0.4 \) [red, Fig. 1(g)] samples.

FIG. 3. (Color online) (a) Symmetrized EDCs for the \( x = 0.4 \) sample at \( k_f \) as a function of the Fermi surface angle \( \phi \), from the node (top) to the antinode (bottom). The cuts are measured along the \( \Gamma - M \) direction. Each curve is offset for clarity. (b) The same as (a) but for the \( x = 0.1 \) sample.
First, we discuss the gap. There are three possible scenarios that explain the difference in the gap size: (I) A scenario where disorder leads to broadening of the band structure features in $x = 0.1$ which hide the gap. However, high-resolution powder x-ray diffraction [24] and NMR experiments [25] indicate that $x = 0.1$ samples are more ordered than $x = 0.4$ ones. (II) A scenario where $\Delta$ opens only below $T_c$. It could be that in our experiment the surface of the $x = 0.4$ sample is below $T_c$, but the $x = 0.1$ surface, is not since its $T_c$ is lower. In this case only the $x = 0.4$ sample will show a gap. The problem with this scenario is the observation of a Fermi arc in $x = 0.4$, which does not exist below $T_c$ in any other cuprate. (III) A scenario where both samples are above $T_c$, but there is an intrinsic difference in their gap size. The problem here is again that in other materials there is no gap above $T_c$ in extreme overdoped samples [26]. Further experiments are needed to clarify this point.

In conclusion, we present ARPES data from CLBLCO. We find that the surface doping is independent of the bulk doping or the Ca to Ba ratio. We also demonstrate that the gap can be measured in this system. The hopping parameter $t$ is larger for $x = 0.4$ than for $x = 0.1$ in the overdoped sides. This suggests that $T_c^{\text{max}}$ is correlated with electron-orbital overlaps on neighboring sites. Note added in proof. Recent NMR measurements indicate that in CLBLCO there is a gap above $T_c$ in extreme overdoped samples [27]. This suggests that there is an intrinsic difference in the anti-nodal gap size between the two families.

This research was supported by the Israeli Science Foundation (ISF) and the joint German-Israeli DIP Project.

\[ \text{Fig. 4.} \text{ (Color online) Main: The MDC peak position, extracted from the nodal dispersion measured along } \Gamma-Y, \text{ as a function of } k \text{ for } x = 0.1 \text{ (black squares) and } x = 0.4 \text{ (red circles). Note the axis breaker. The solid lines are a linear fit to the data in the } -150 < E < -50 \text{ meV range from which the high-energy velocities } V_{HE} \text{ are extracted. Inset: A histogram of high-energy velocities obtained from a series of CLBLCO samples with } x = 0.1 \text{ (black, filled) and } x = 0.4 \text{ (red, crossed).} \]

\[ -150 < E < -50 \text{ meV. The transition between these regimes is known as the kink and involves the electron dispersion renormalized due to correlations [18] or coupling between electrons and low-energy bosonic degrees of freedom [19]. The slope of the dispersion } \partial V_{HE}/\partial k \text{ provides the velocity in the low- } (V_F) \text{ and high-energy } (V_{HE}) \text{ regimes. The results are similar to other overdoped materials [20].} \]

We did not find differences with statistical significance in $v_F$ between samples with different $x$. This is in agreement with previous work [20]. As for $v_{HE}$, the results are summarized in the inset of Fig. 4 as histograms. For the $x = 0.1$ family, the average high-energy velocity is $V_{HE} = 1.53(04) \text{ eV } \AA$, while for the $x = 0.4$ it is $V_{HE} = 1.73(04) \text{ eV } \AA$. Despite the velocity distribution overlap, the average velocities differ by $3.5\sigma$, and hence are statistically different with 99.5% confidence. Using these velocities we can now calculate all the TB coefficients by $\partial V_{HE}/\partial k = 4at \sin(k_{node}a)$. The unit cell parameter $a = 3.91 \text{ Å}$ is nearly family independent [5]. The coefficients are presented in the Supplemental Material [21] and are in agreement with previously published values [22,23].

From the data presented, we can draw several conclusions. First, we discuss the ratio of velocities $((V_{HE}^{0.4})/(V_{HE}^{0.1}))^2 \simeq 1.26 \pm 0.08$. Despite the large error bar, this ratio is very close to that expected from the ratio of the superexchange $J$ between families. This ratio is given by $J(0.4)/J(0.1) = \frac{T_c^{\text{max}}(0.4)}{T_c^{\text{max}}(0.1)} \simeq 1.4$ (see Fig. 1). Therefore, the $J \propto t^2$ relation is obeyed, and $T_c^{\text{max}}$ depends on orbital overlaps even when the measurements are done in the doped phase.

However, ARPES measurements do not necessarily represent the bulk properties. For example, the buckling angle might change close to the surface. Nevertheless, if this happens in CLBLCO, it might affect both families equally. The fact that the ratio of $J$ measured magnetically agrees with the ratio of $t^2$ measure by ARPES supports this notion.


