Proximity, magnetoresistance & gating effects in bilayers (BL) of doped topological Bi₂Se₃ & superconducting NbN

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- Part I compares bilayers to bare NbN films
 <u>Supercond. Sci. Technol.</u> 28 025003 (2015) & <u>http://arxiv.org/pdf/1409.2975v1.pdf</u> (2014)
- Part II Magnetoresistance & gating effects on these bilayers & films http://arxiv.org/pdf/1506.08584v1.pdf (2015)

Outline

- Some background
- Ultra-thin bilayers of Bi₂Se₃-NbN for studying topological superconductivity

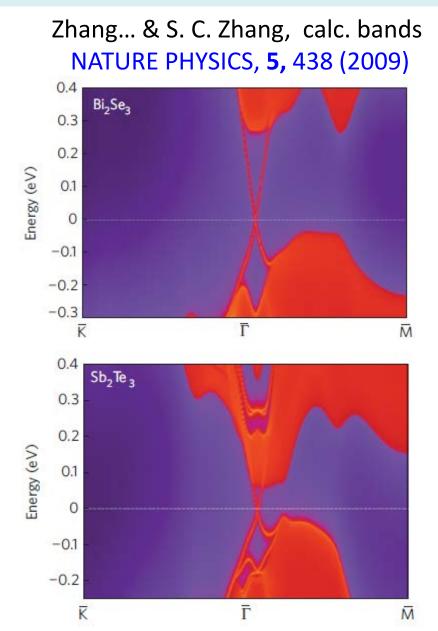
Part I

• Proximity effects in these bilayers

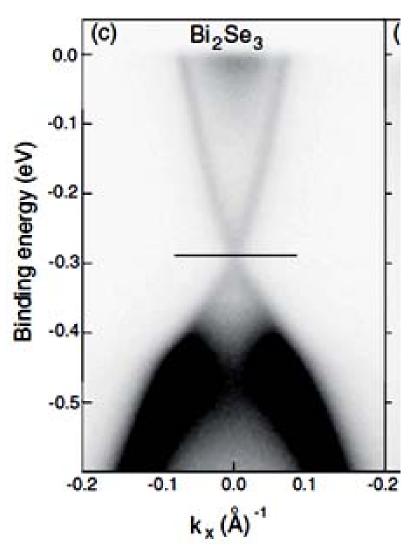
Part II

- Magnetoresistance & gating effects in Bi₂Se₃ films
- Magnetoresistance & gating effects in the bilayers
- Interpretation of the results in terms of vortex physics & pinning
- Alternative interpretations

Topological insulators are bulk insulators with surface conductance & a single Dirac cone of the surface states



T. Kirzhner &... *A. Kanigel,* meas. ARPES, PRB 86, 064517 (2012)



Motivation for studying topological superconductors (TSC)

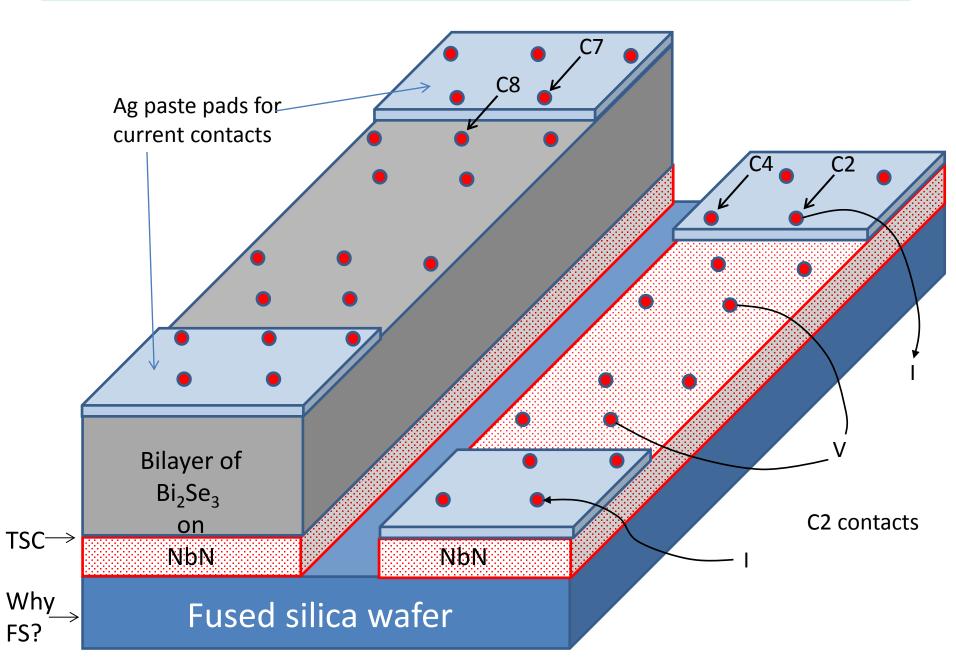
- It is predicted that Majorana fermions (MFs) exist in the vortex cores of topological superconductors (TSC)
- These MFs will appear as zero energy modes (or ZBCP) in conductance spectra of TSC/N junctions (N is a normal metal)
- They should be robust against disorder & decoherence
- & thus might be useful in quantum computing (*)

(*) Fermionic Quantum Computation Sergey B. Bravyi and Alexei Yu. Kitaev, Annals of Physics **298**, 210–226 (2002)

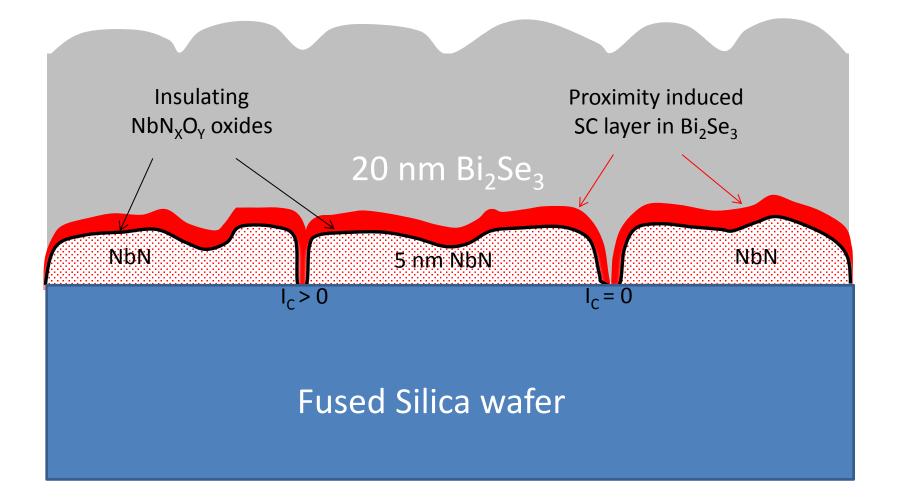
Why studying transport in ultra-thin bilayers of Bi₂Se₃-NbN?

- If the NbN film is too thick (>10nm) & T<T_c, it will short all the effects of the Bi_2Se_3 cap-layer \rightarrow no TSC at the interface could be investigated.
- Thus a thin NbN islands layer (~3 to 5nm) is needed, that behaves as a network of weak-links when bridged by the Bi₂Se₃ cap-layer
- → current will have to flow via the interface with the Bi₂Se₃ layer, allowing for the TSC to be studied
- Percolation paths of weak-links will constitutes 1D TSC, and their geometry relative to the voltage contacts will determine the TSC transport properties in the bilayers
- In the following slides, this will be explained in details
- Some relevant parameters for 3-5nm thick NbN films: Tc ~ 5-8 K, ξ ~ 5 nm, λ ~ 500 nm, Δ ~ 1.5 meV A. Kamlapure et al. Appl. Phys. Lett. **96**, 072509 (2010)

A scheme of the film, bilayer & contacts geometry on the wafer

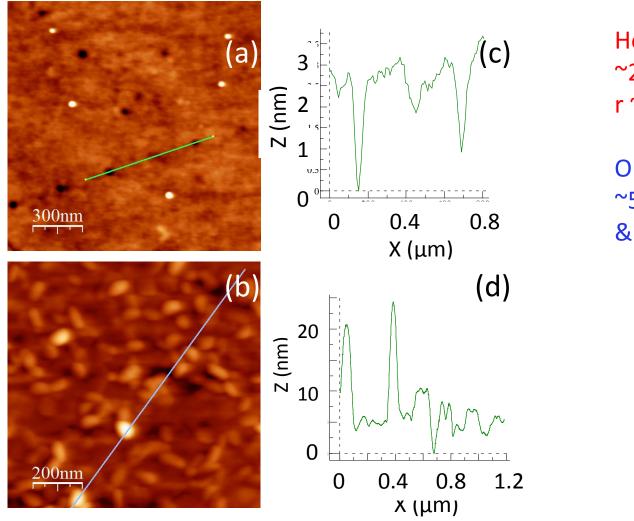


A model for the transport in the bilayer



- Weak-links between the NbN grains (I_c with serial resistance)
- Proximity induced superconductivity in Bi₂Se₃ cap layer (at interface)

Topography of: (a) a 5nm thick NbN film, and of a bilayer (b) of 20nm Bi₂Se₃ on 5nm NbN on FS & typical line profiles

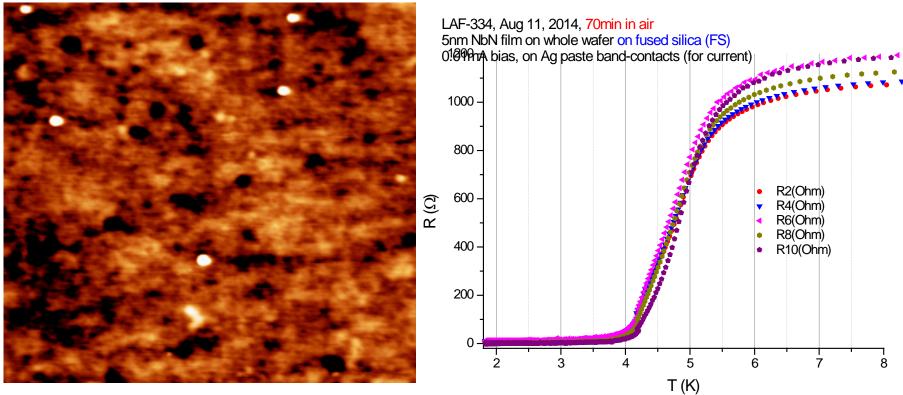


Holes: ~2-3 nm deep r ~ 30 nm

Outgrowth: ~5-7 nm high & similar r

On $SrTiO_3$ the NbN films are so smooth, that there are no weak-links.

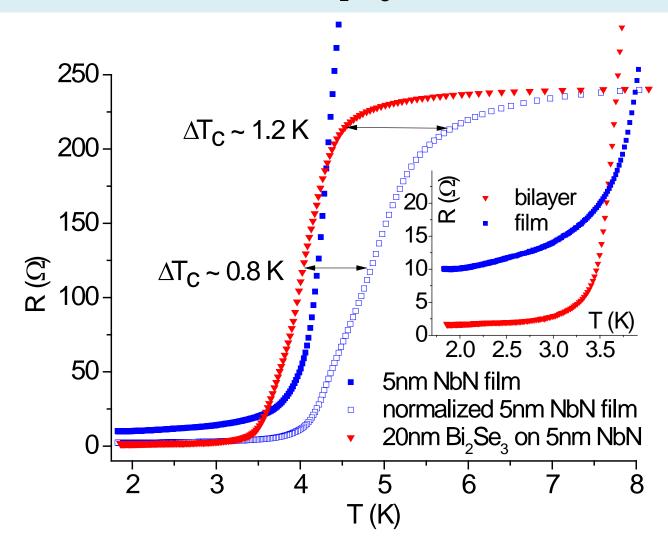
RT & contrast-topo of a 5nm thick NbN film on FS



1.6 x 1.5 μm²

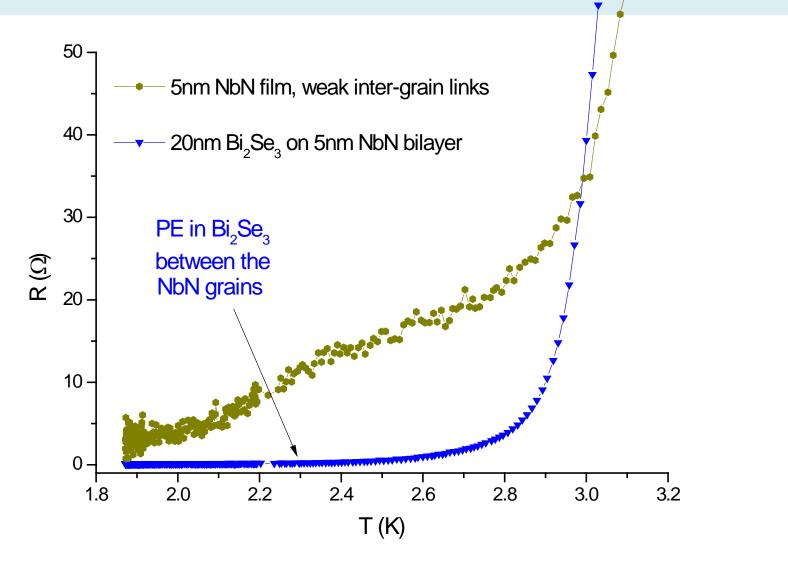
- NbN islands (bright areas) connected by weak-links (dark areas)
- rms roughness: ~0.5nm (0.1nm on STO not shown)

R vs T of a 5nm thick NbN film after 10min in air, compared to an *in-situ* deposited bilayer of 20nm Bi₂Se₃ on 5nm NbN (LAF-BL-334)



Proximity effect (PE) in BL suppresses superconductivity in NbN grains (lower T_c)

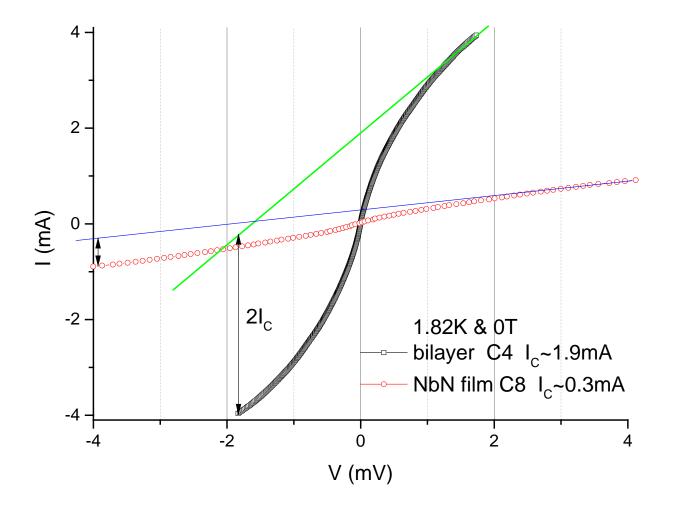
Similar wafer (LAF-BL-333): Below 3K & without normalization of R



• Induced SC in Bi₂Se₃ at the interface with NbN with no serial resistance

• T_c (bilayer at R=0) ~ 2.3K, No T_c (film) down to 1.9K \rightarrow inverse PE in BL...

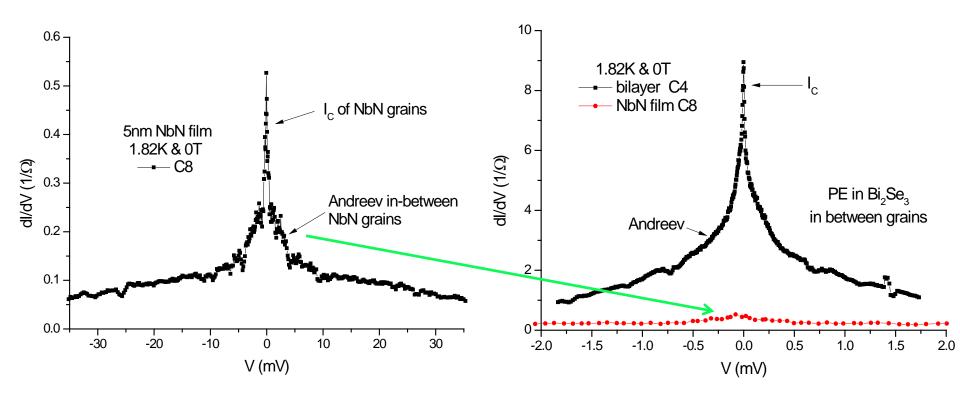
Same wafer (LAF-BL-333): IVC at 1.82K



Both have I_c at 1.82K, but I_c(bilayer) is about 6.3 times larger than I_c(film)

I_c enhancement in BL at low T indicates (inverse) PE in the TI

Conductance spectra of in-situ prepared bilayer and film (LAF-BL-333)

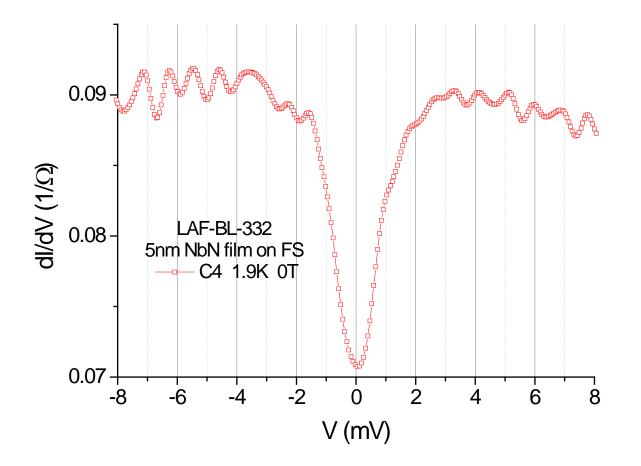


- I_c and Andreev peaks
- Andreev enhancement in BL at low T indicates (inverse) PE in the TI
- ZBC (1/R) of BL is about 20 times higher than that of the NbN film [R(normal of NbN film) ~ 6 x R(normal bilayer)]

Scattering of the data (LAF-BL-332) after 17min exposure to ambient air: two extreme cases for the 5nm NbN film 30 -Weak & strong-links between NbN grains 25 BL: bilayer of 20nm Bi₂Se₃ on 5nm NbN 20 -F: 5nm NbN film 15 R (<u>C</u>) BL C6 BL C7 No T_{c} (R=0) of film at C4 BL C8 10 **BL C10** FC2 T_c(R=0) of BL~2.2-2.3K FC4 5 T_{c} (R=0) of film at C2 ~ 2.4K 0 3.0 2.0 2.5 3.5 T (K)

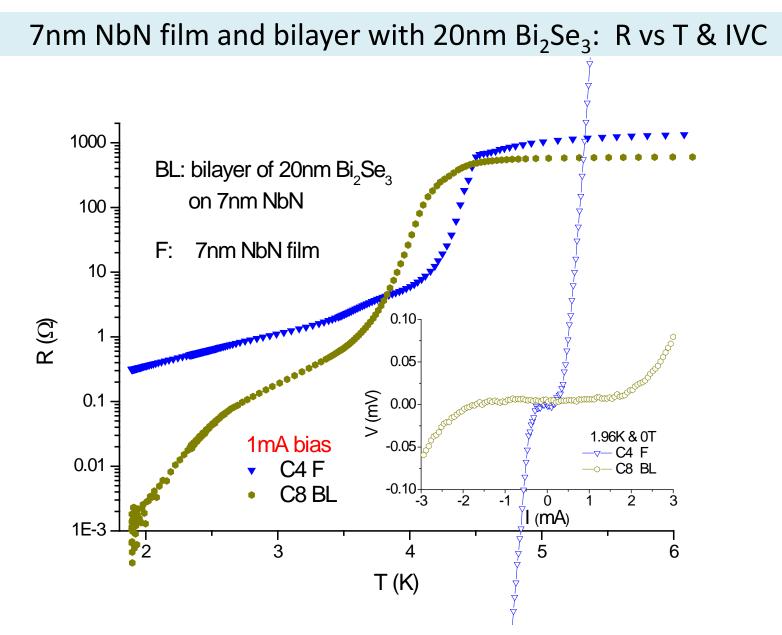
- Percolation of strong-links between the SC grains that connects the V-contacts of C2
- Percolation of weak-links between the SC grains that cuts the V-contacts of C4
- PE in Bi_2Se_3 in the BL, T_c of 2.2-2.3K, low scattering of the data (thick cap layer)

Tunneling conductance of the C4 contact of LAF-BL-332



• Gap-like feature at $2\Delta \sim 2 \text{ mV} \rightarrow \Delta$ of the NbN grains $\sim 1 \text{mV}$ (assuming 1 SNS junction)

• The weak-link here could originate in stronger defects like nm scratches etc.

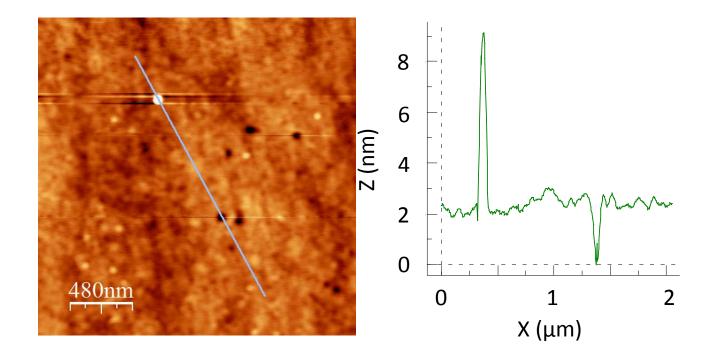


- Stronger links in thicker film which is less resistive at low T
- Need a higher bias to see PE in Bi₂Se₃ below 3.5K, see IVC at low T.

Conclusions I

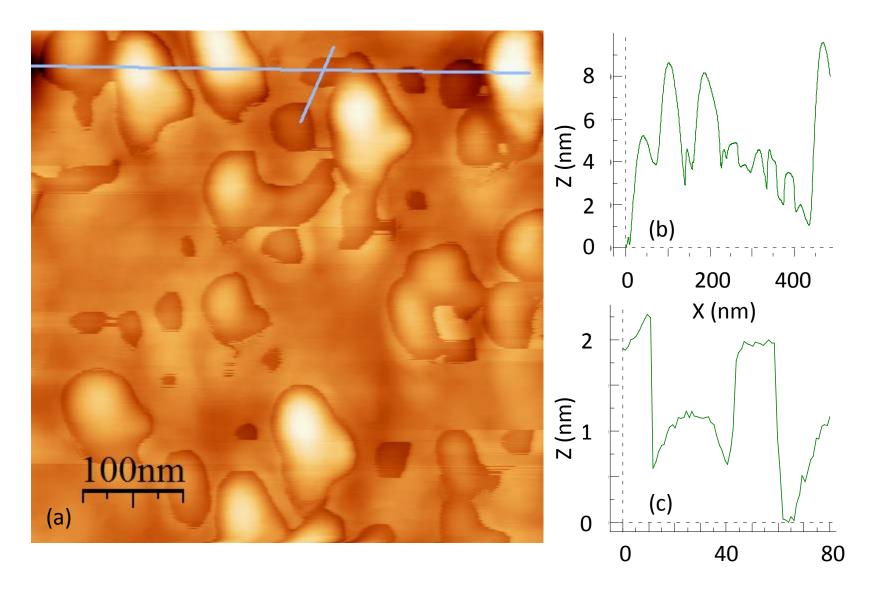
- Transport properties of *in-situ* prepared bilayers of a TI (Bi₂Se₃) & a SC (NbN) were compared to the bare SC film
- Standard proximity effect (PE) was observed in all bilayers in the main part of the SC transition (high T), where the TI layer (N) suppressed T_C of the SC layer
- At low T, an inverse PE in the TI of the bilayers was observed, as indicated by enhancement of I_c & Andreev conductance in the bilayer compared to the SC film
- Next we shall present <u>part II</u> which deals with <u>gating</u> effects on the magnetoresistance of even thinner bilayers and films.

AFM image of a 3nm thick NbN layer on FS



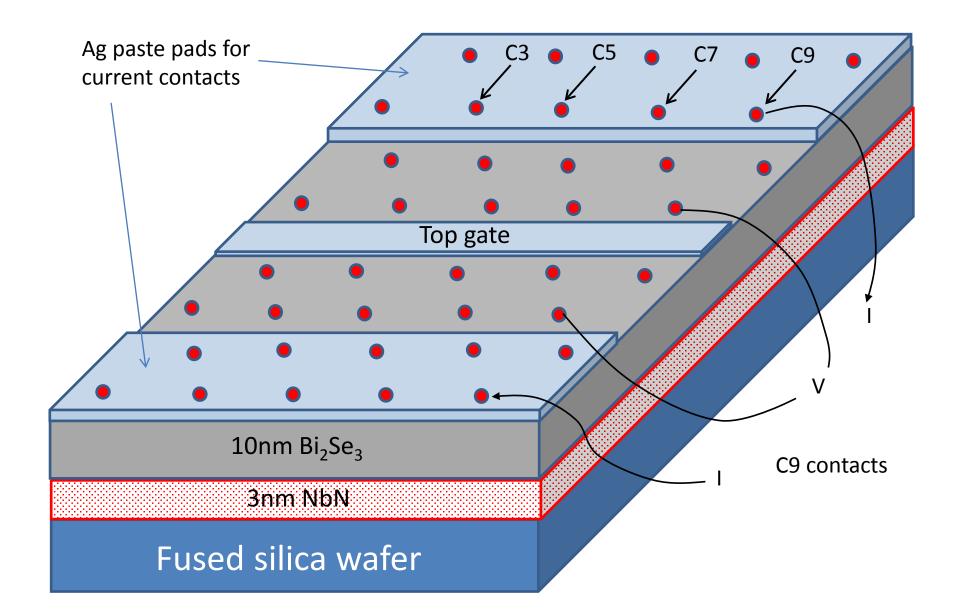
- Nano holes of ~70nm diameter
- ~0.35 nm rms roughness

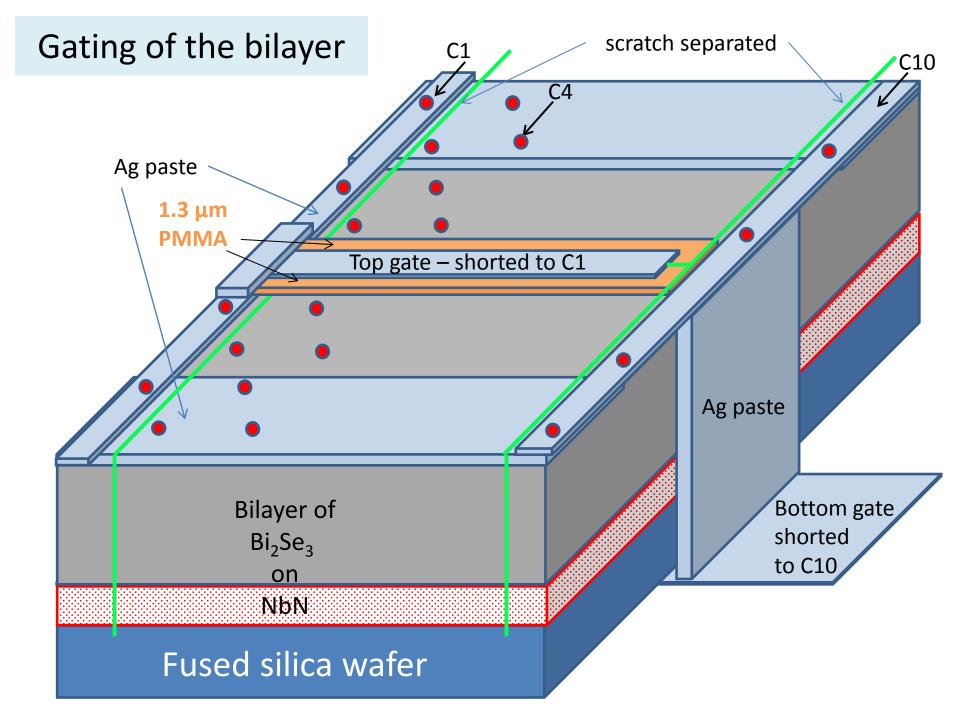
AFM image of a 10nm Bi_2Se_3 on 3nm NbN bilayer on FS & the corresponding line profiles in (b) & (c)



~1.1 nm rms roughness & Quintuple steps of 1 and 2 nm can be seen in (c)

A schematic drawing of the bilayer & contacts geometry now

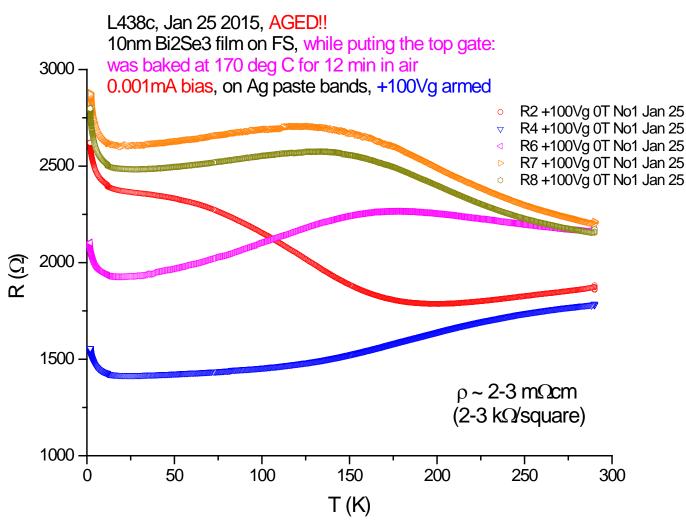




10 nm Bi₂Se₃ reference film on fused silica (1) R vs T results

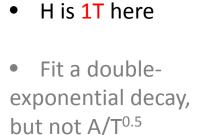
For similar resistivity, The electron densities according to Butch *et al.* PRB **81**, 241301R (2010) are:

- 4-6 x 10¹⁷ elect./cc From Hall measurements
- 3-4 x 10¹⁷ elect./cc From SdH oscillations

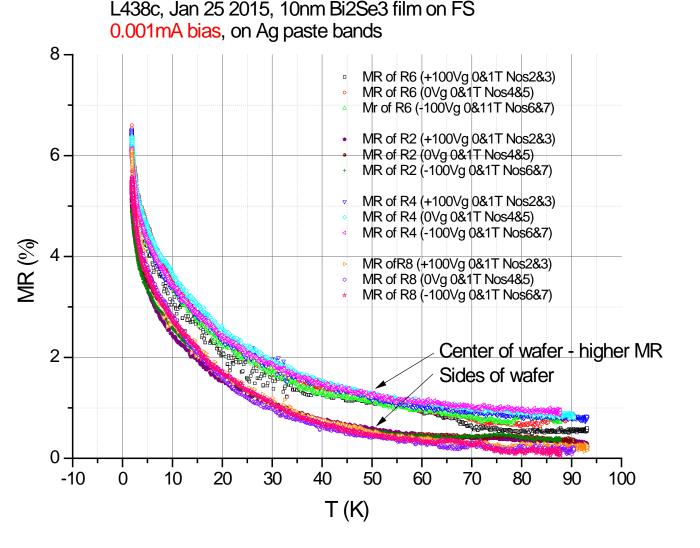


10 nm Bi₂Se₃ reference film on fused silica (2) MR results

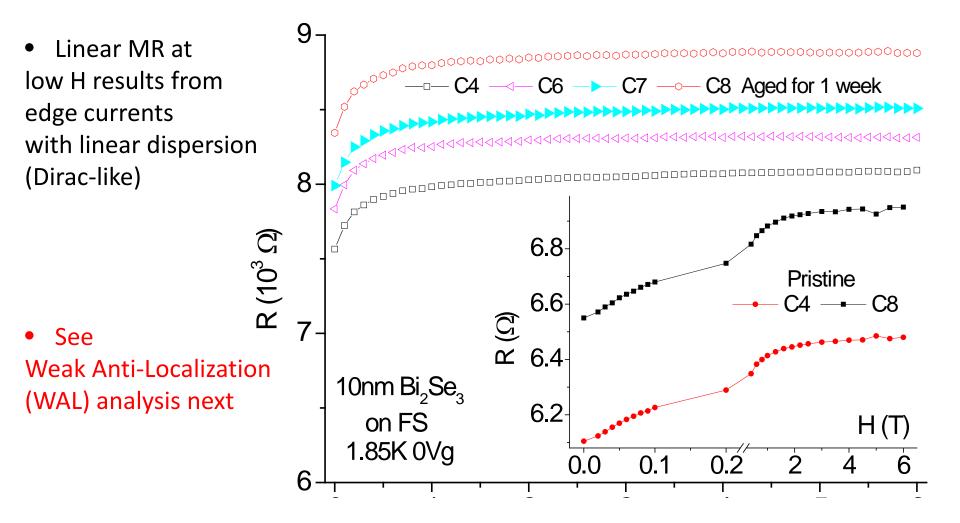
• Magnetoresistance MR = [R(H)-R(0)] / [R(H)+R(0)]/2 H is the magnetic field



- Almost insensitive to gate-voltage Vg, unlike Steinberg et al.
 PRB 84, 233101 (2011)
- Eg-field of top gate for -100Vg & ε~2.6 is: 2.6x100V/0.00013cm
 =2 MV/cm (vs x 40 higher of Steinberg)

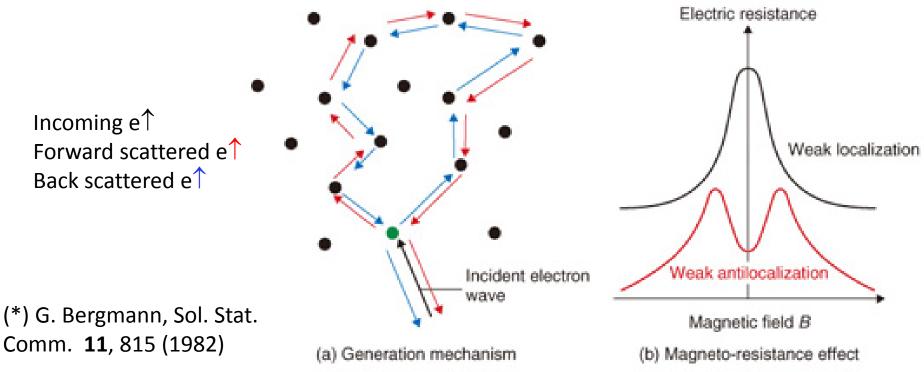


(3) R versus H of another 10nm Bi_2Se_3 film on FS



Weak localization (WL) and Anti-localization (WAL)

- Disordered 2D electron systems can have coherent closed-loop electron trajectories due to elastic scattering (a), which lead to WL when interfere constructively, and WAL when interfere destructively.
- Since MR \propto dR/dH & added flux destroys pure WL and WAL, MR<0 for WL & MR>0 for WAL, see (b)
- Under strong spin-orbit int., the spins rotate & $\psi(e^{\uparrow})$ returns to itself only after 2 loops yielding added phase of $\Delta \phi = 4\pi$.
- For 1 loop $\Delta \phi = 2\pi$, the spins $\uparrow \& \uparrow$ rotate in opposite directions & one can show that they interfere destructively, thus WAL occurs (*).



Weak anti-localization – WAL & quantum diffusion The Hikami-Larkin-Nagaoka (HLN) model (*)

$$\Delta G_{\Box}(B) \cong \alpha \frac{e^2}{\pi h} \left[\psi(\frac{1}{2} + \frac{B_{\phi}}{B}) - \ln(\frac{B_{\phi}}{B}) \right]$$

ψ is the Digamma function d/dz[ln Γ(z)] & for z=n, Γ=(n-1)! $B_{\phi} = \hbar / 4eL_{\phi}^2 \qquad \& \qquad \alpha \approx -1/2$ $\Delta G_{\Box}(B) = G_{\Box}(B) - G_{\Box}(B=0) \qquad \& \qquad G_{\Box}(B) = L/WR_{\Box xx}$

For B=1T & phase coherence length L_{ϕ} in nm:

$$1.2 \times 10^5 \frac{R(B) - R(0)}{R(0)R(B)} \cong \psi \left(\frac{1}{2} + \frac{156}{L_{\phi}^2}\right) - \ln\left(\frac{156}{L_{\phi}^2}\right)$$

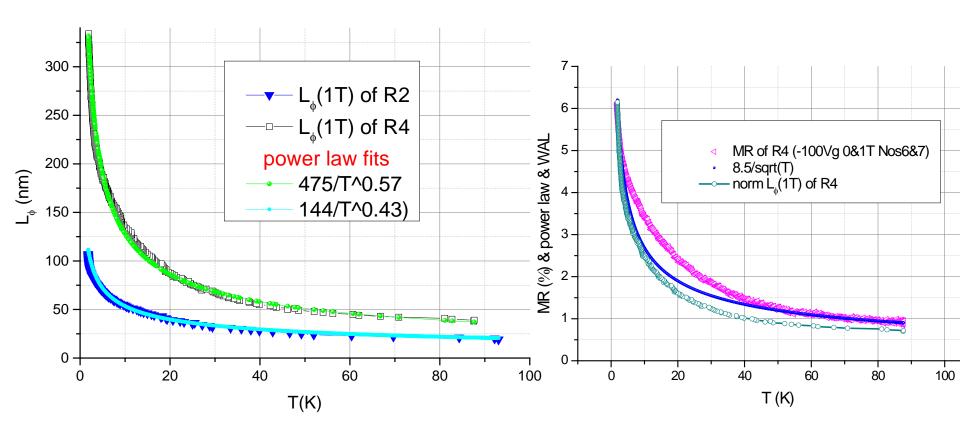
- R(B)-R(0) is the un-normalized magnetoresistance (MR)
- Since R is temperature dependent, this formula yields $L_{\phi}(T)$

(*) Prog. Theor. Phys. 63, 707 (1980)

+ spin-orbit => Interference effects 279^{3} 8^{3} 4^{3} 5^{5} 6^{5} 5^{5} 6^{5} 3^{4} 7^{7} 10^{2} 2^{2} 8^{3} 8^{3} 7^{7}

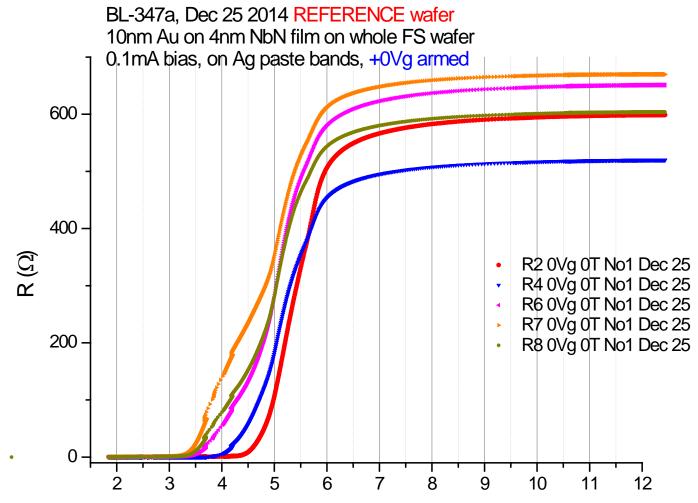
2D + disorder

10 nm Bi₂Se₃ reference film on fused silica (4) MR, WAL & power law results vs T



- On the same wafer, different L_{ϕ} & power-law exponents for different contacts
- On same contact, different behavior of MR, power-low (-1/2 for 2D) & L_{ϕ} (WAL)
- $L_{\phi} = \sqrt{D\tau_i}$ where for T>2K, τ_i is the electron-phonon scattering time (shortens vs T)

(1) A second reference bilayer of 10 nm gold (Au) on 4nm NbN on fused silica R vs T results

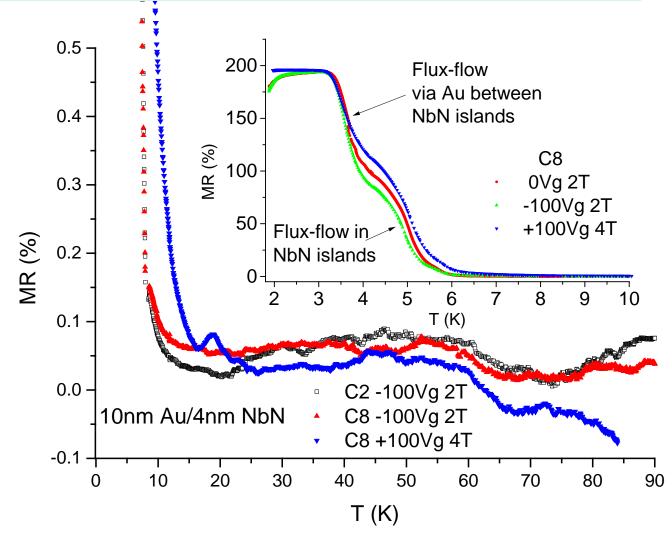


T /I/\

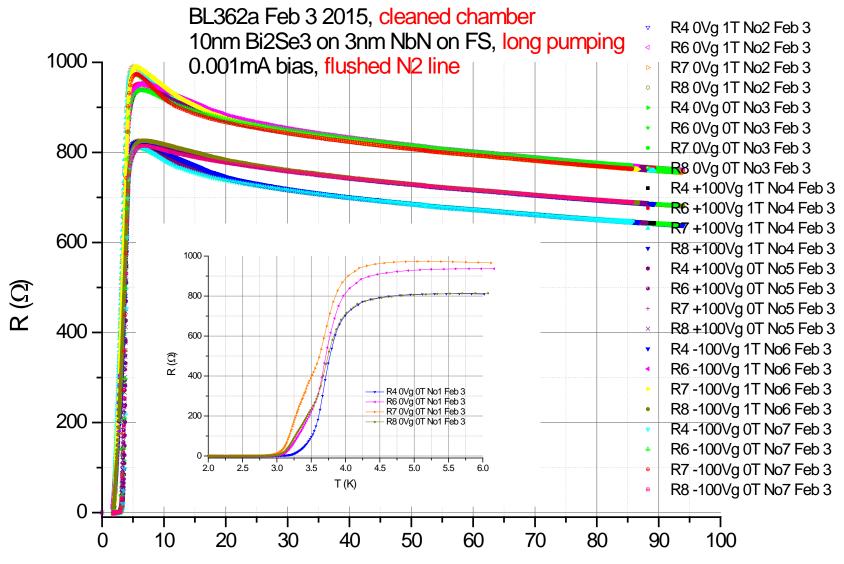
(2) MR results & Flux-flow (FF) in this bilayer below Tc(10 nm gold (Au) on 4nm NbN on fused silica)

- H is 2 & 4T here
- MR(T<Tc) due to flux-flow (FF) of NbN grains and in between the grains in the Bi₂Se₃ cap-layer
- For 2T, negligible
 MR of ~± 0.1% above
 Tc ~ 7-8 K

 At 4T, the SC MR starts at ~12 K due to FF(4T) > FF(2T)



(1) R vs T of a bilayer of 10 nm Bi₂Se₃ on 3nm NbN on fused silica [low R_{max} case]

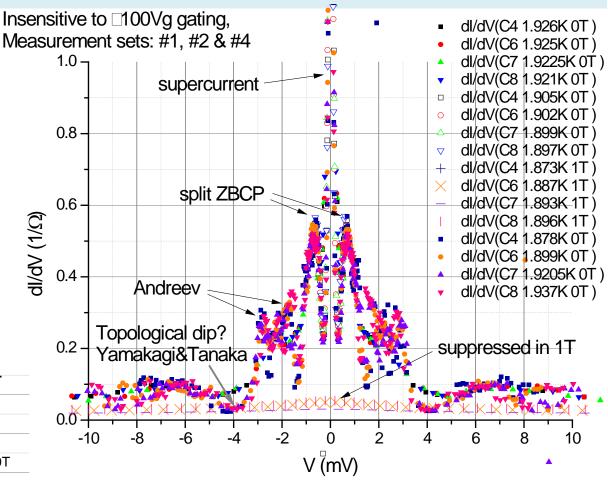


(2) dI/dV & IVC of BL362a (10 nm Bi_2Se_3 on 3nm NbN on FS)

- Vg=0V
- I_C(1.9K)~1.2 mA (strong links)
- dI/dV is almost fully suppressed at 1T



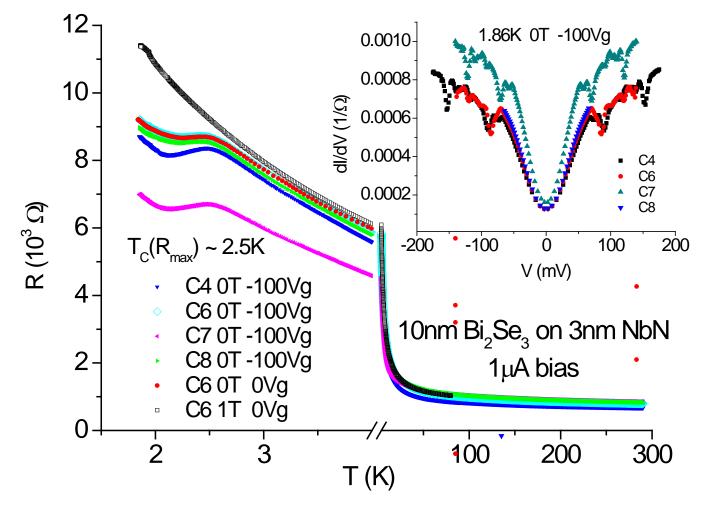
V (mV)



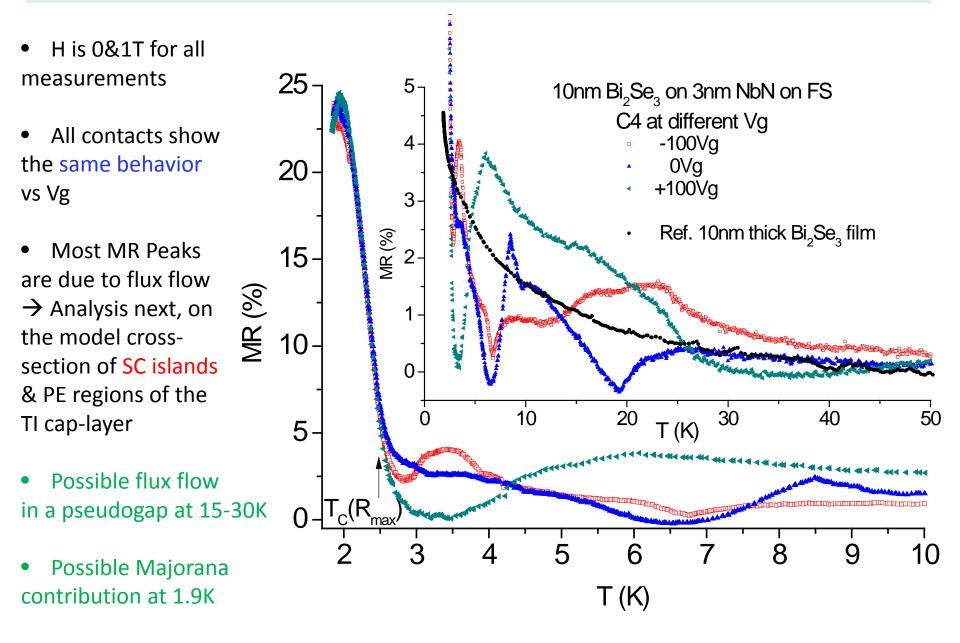
Monotonous decreasing MR vs temperature (not shown)

(1) R vs T of another bilayer of 10 nm Bi₂Se₃ on 3nm NbN on fused silica [highest resistance, some air in NbN deposition?]

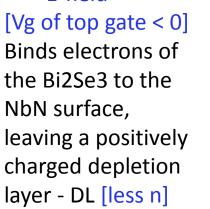
- Semiconducting
- Very weak SC suppressed under 1T
- Tunneling between
 NbN grains



(2) MR of this bilayer of 10 nm Bi₂Se₃ on 3nm NbN on FS

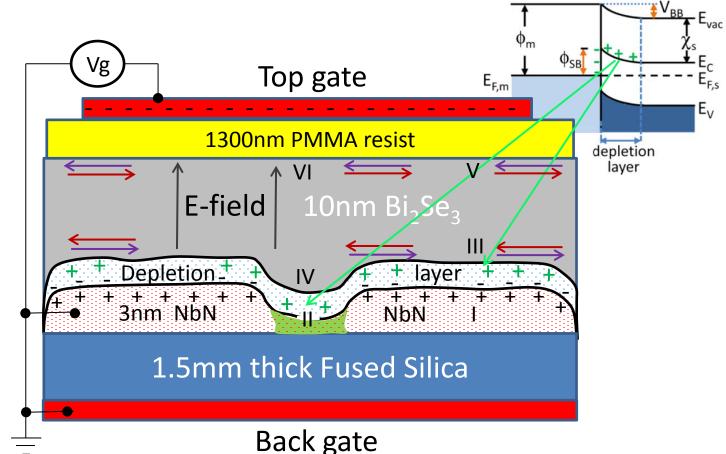


Model: electron density (n) under E-field in the bilayer



+E-field

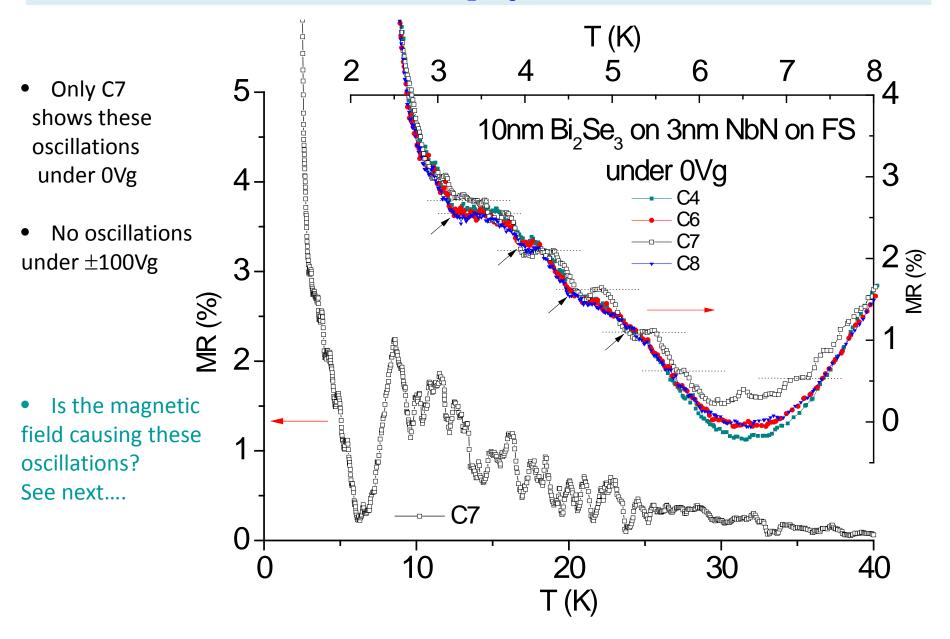
 -E-field binds holes, leaving electrons in the DL



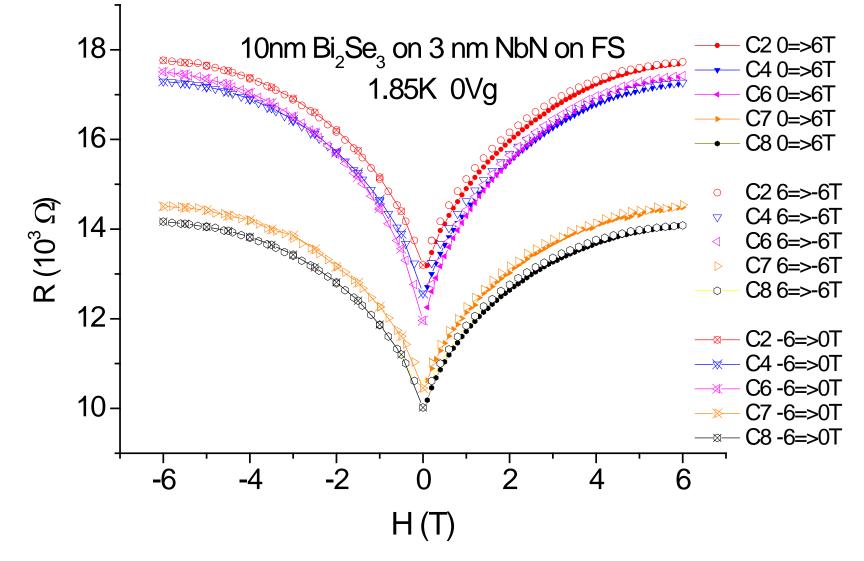
MR peaks analysis:

- For OVg: MR peaks at 7-15K, due to NbN islands regions I, and PE in region III MR knee at 2.5-6K is due to PE in region II (& IV ?) Large MR peak below 2.5 K can be due to enhanced PR in region IV or to other effects
- For -100Vg: MR peak at 3-4K originates in the –e depleted region II
- For +100Vg: Broad MR peak at 6K originates in -e rich region II

(3) MR of C7 of this 10 nm Bi₂Se₃ on 3nm NbN on FS bilayer



(4) R vs H of this 10 nm Bi₂Se₃ on 3nm NbN on FS bilayer



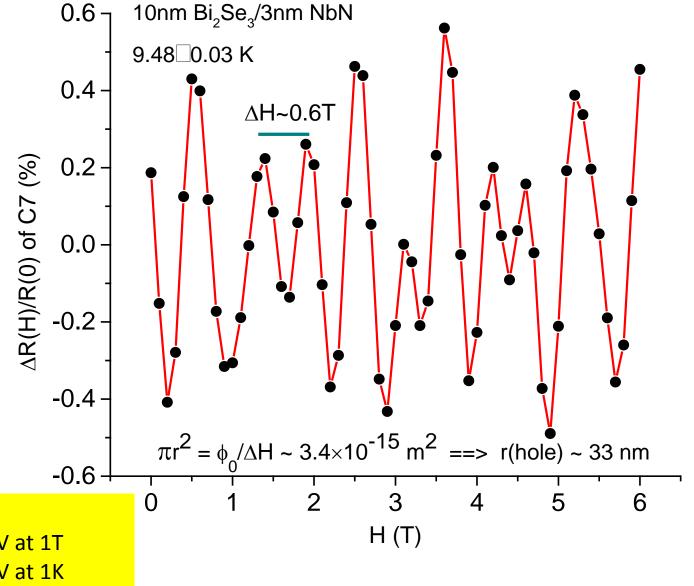
No oscillations here, but see next...

(5) $\Delta R/R(0)$ vs H at 9.5K of this bilayer (10 nm Bi₂Se₃ on 3nm NbN)

- $\Delta R(H)$ is R(H) minus a smooth background
- Oscillations vs H with a period of $\Delta H \sim 0.6T$ adding ϕ_0 for each period of $\Delta H \sim 0.6T$
- Corresponding to a hole radius of r ~ 33nm
- In agreement with the AFM image
- Tc(islands) \geq 10K

Similar energy scales:

Magnetic $\sim \mu$ H ~ 0.06 meV at 1T Thermal $\sim k$ T ~ 0.08 meV at 1K Δ E(Landau at 1T) = $h\omega_c/2\pi = 0.12$ meV



Conclusions II

- MR & WAL of 10nm Bi₂Se₃ films were observed in agreement with the literature
- MR of the 10nm Bi₂Se₃ on 3nm NbN bilayers was:
 - 1. Strongly dependent on the gate voltage Vg
 - 2. Varied with the peak resistance at low T
 - 3. Showed Vg dependent peaks and dips structure
 - 4. Oscillations vs H at 9.5K are

due to flux quantization in nano holes

& indicate Tc above 10K of the SC islands

5. MR vs T response is due mostly to vortex physics & pinning

6. Enhanced PE in the Bi_2Se_3 in between NbN islands at T=1.9 K

can be due to: helical surface currents contribution,

or to Majorana zero mode contribution